



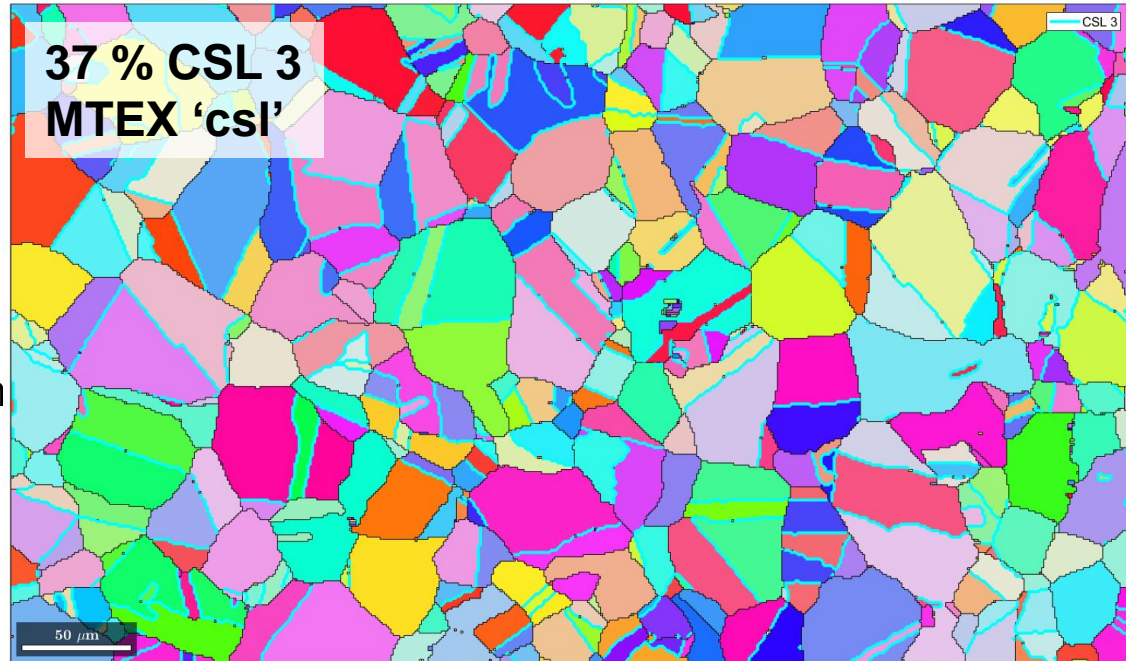
# Annealing twins in reconstructed austenite

Tuomo Nyysönen

Senior researcher, [tuomo.nyysonen@swerim.se](mailto:tuomo.nyysonen@swerim.se)

# CSL 3 annealing twins

- Found in austenitic metals and alloys
- CSL 3 corresponds to
  - $60^\circ$  around  $\{111\}$
- At least one boundary plane in an annealing twin parallel with the CSL plane
- Grain boundaries faceted and angular in shape



# Annealing twins in low-carbon steels

- At a high temperature, austenite undergoes a nucleation and growth process before the prior austenite structure is finished
- During growth, CSL 3 type twin boundaries nucleate at grain boundaries during the boundary migration steps
- Annealing twins in austenite...
  - Affect the nucleation and grain growth of austenite
  - Affect mechanical properties
- Necessary to characterize, just as important as (if not more than) other grain boundaries

# Difficulties with finding PAG annealing twins

- Typically not visible in LOM, even when etching for PAG boundaries
  - Etching techniques based on segregation of impurities to high-energy grain boundaries
  - Twins have a high coherency and thus very low energy!
- What about **PAG reconstruction with EBSD?**



Source: Vander Voort,  
<http://www.georgevandervoort.com/revealing-prior-austenite-grain-boundaries-in-heat-treated-steels-article/>

# Example: MCL route

```

% load the data
mtexdata martensite

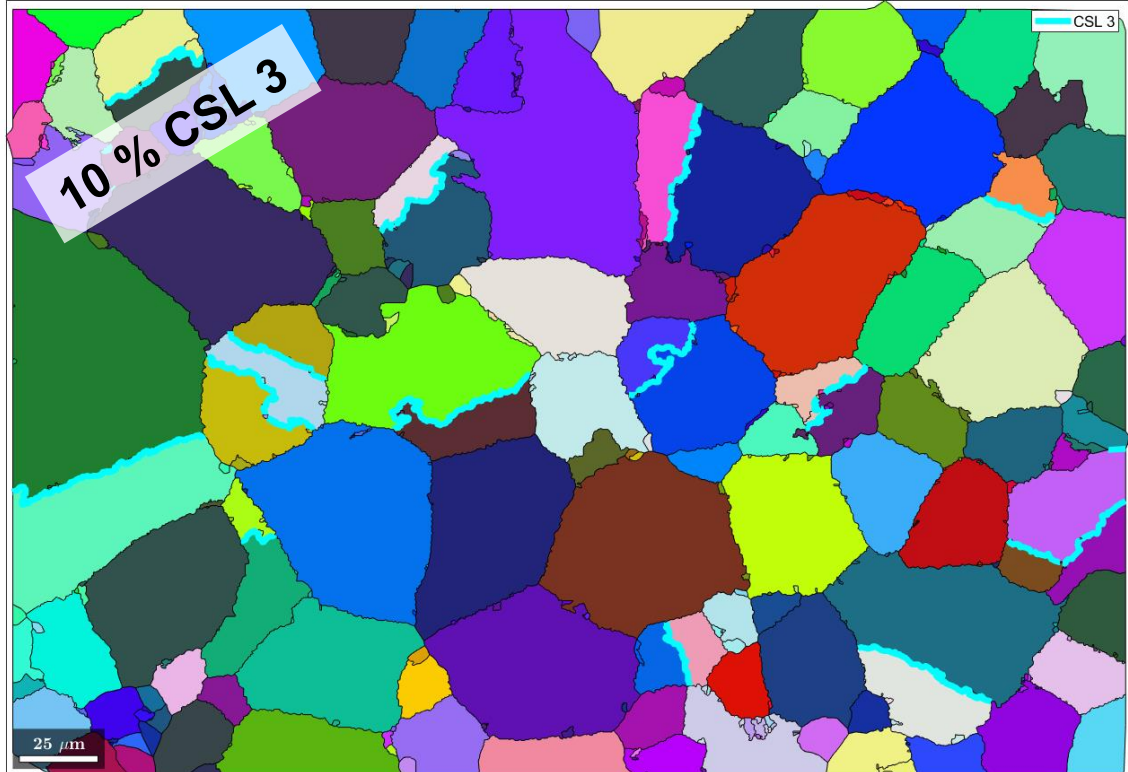
% grain reconstruction
[grains,ebstd.grainId] =
calcGrains(ebstd('indexed'),'angle',3*degree);
ebstd(grains(grains.grainSize < 3)).rotations =
quaternion.nan;
ebstd(grains(grains.grainSize < 3)).phase = 0;
ebstd(grains(grains.grainSize < 3)).grainId = 0;
[grains,ebstd.grainId] =
calcGrains(ebstd('indexed'),'angle',3*degree);
%%
job = parentGrainReconstructor(ebstd,grains) % define job
job.calcParent2Child; % get representative OR

%%
[fit,c2cPairs] = job.calcGBFit;
[gB,pairId] =
job.grains.boundary.selectByGrainId(c2cPairs);
plot(gB, fit(pairId));
mtexColorMap white2black

%%
job.calcGraph
job.clusterGraph
job.calcParentFromGraph
job.revert(job.grains.fit/degree > 5)
job.revert(job.grains.clusterSize < 15)

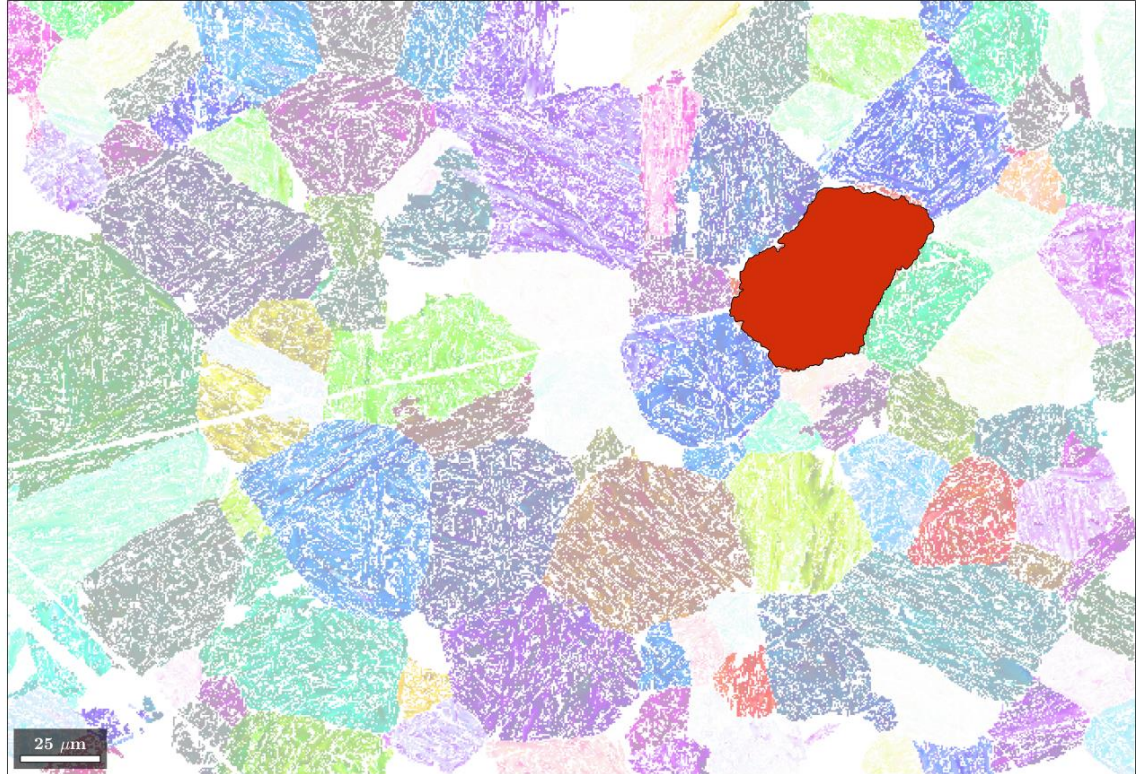
for k = 1:3 % do this three times
job.calcGBVotes('noC2C');
job.calcParentFromVote('minFit',7.5*degree)
end

```



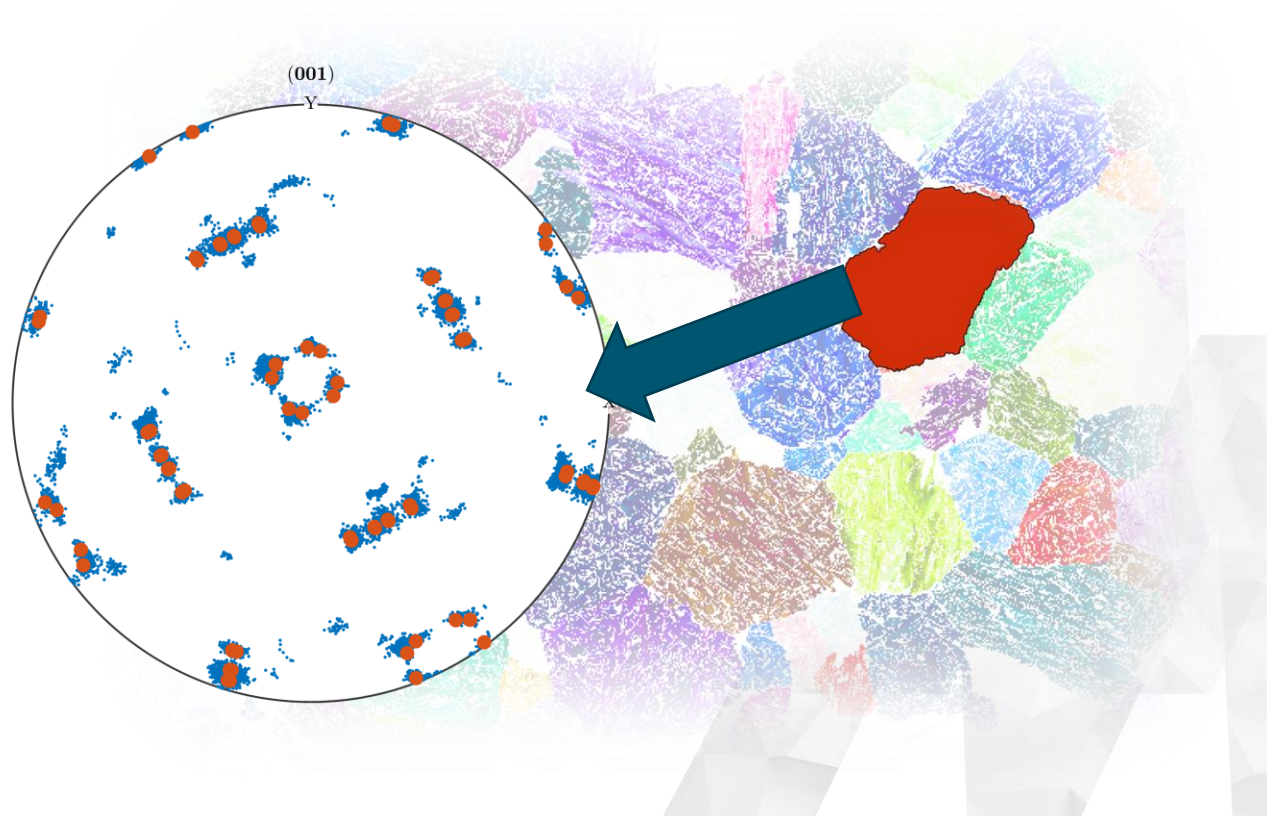
# Were all the twins really indexed?

- Some twin boundaries have been identified
- Some suspicious reconstructions, especially at grain boundaries...
- Let's pick a grain and investigate



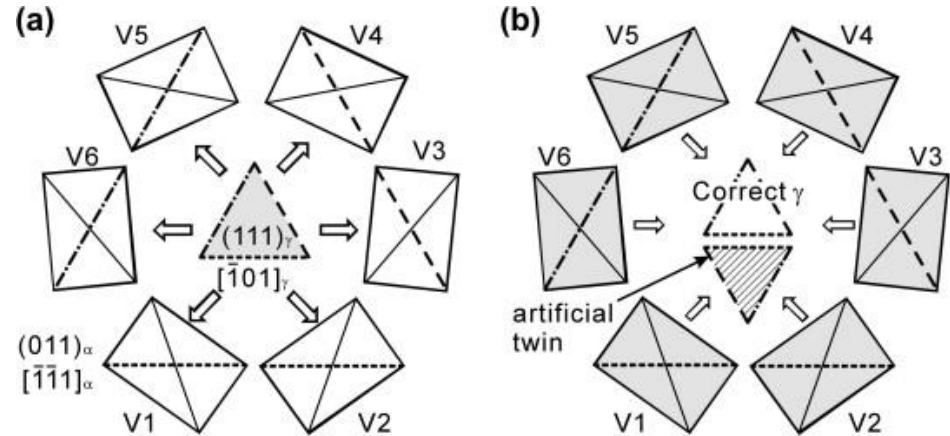
# Were all twins really indexed?

- It would appear that some orientations fall way outside the calculated variants for the determined prior orientation!



# What's the problem?

- K-S orientation relationship:
  - $(111)_\gamma \parallel (011)_\alpha$
  - $[10\bar{1}]_\gamma \parallel [11\bar{1}]_\alpha$
- Laths form on the  $(111)_\gamma$
- Annealing twins share a  $(111)$  plane
  - Variants forming on this particular plane will be identical
- **Annealing twins basically share a packet, assuming K-S OR!**

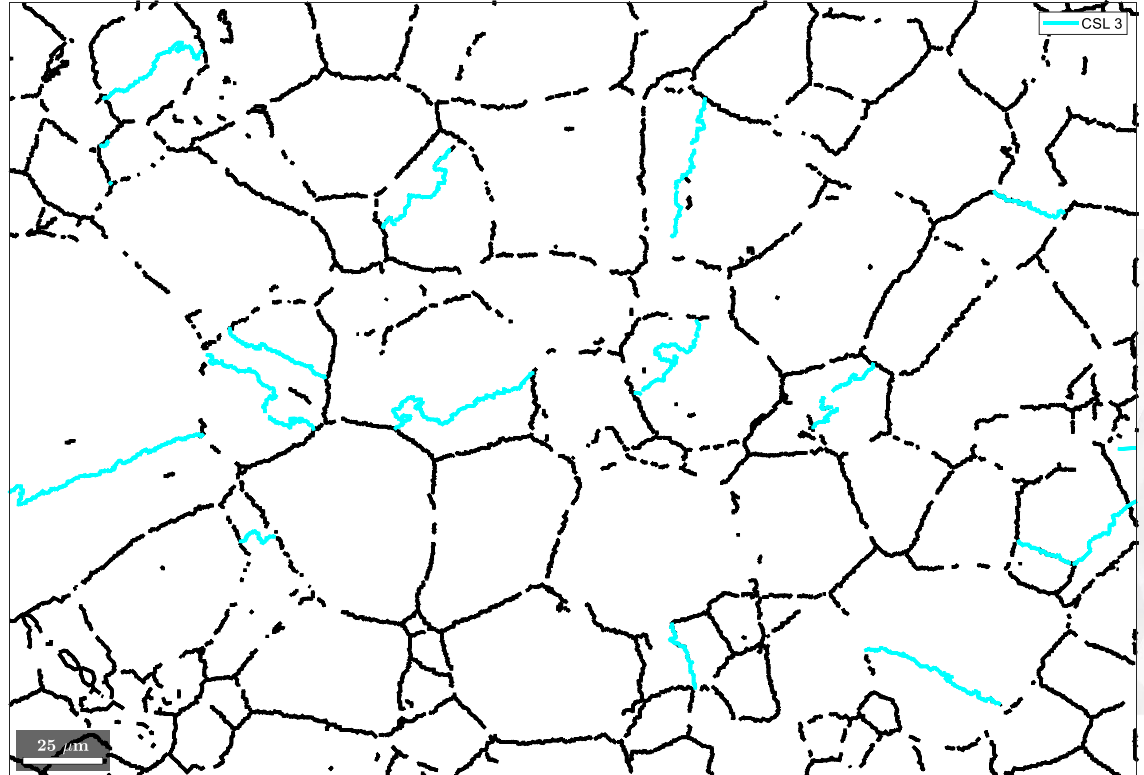


G. Miyamoto et al., Acta Materialia, Volume 58, Issue 19, 2010, <https://doi.org/10.1016/j.actamat.2010.08.001>.



# How does this relate to MCL clustering?

- Variant formation typically aims to accommodate plastic deformation during transformation
- To achieve this, variants from the same packet tend to form next to each other
  - This also occurs across annealing twin boundaries
- **Result: twin boundaries cannot be characterized for MCL!**



# Fortunately, MTEX has a lot of tools available.

- The confidence in a correct prior austenite orientation increases when several neighbors can agree on a common solution
- Kerner + growth approach:
  1. Determine prior orientations that agree with 4 neighbors
  2. Use the reliable orientations as kernels, which are then grown into full austenite grains



# MCL route...

```

% load the data
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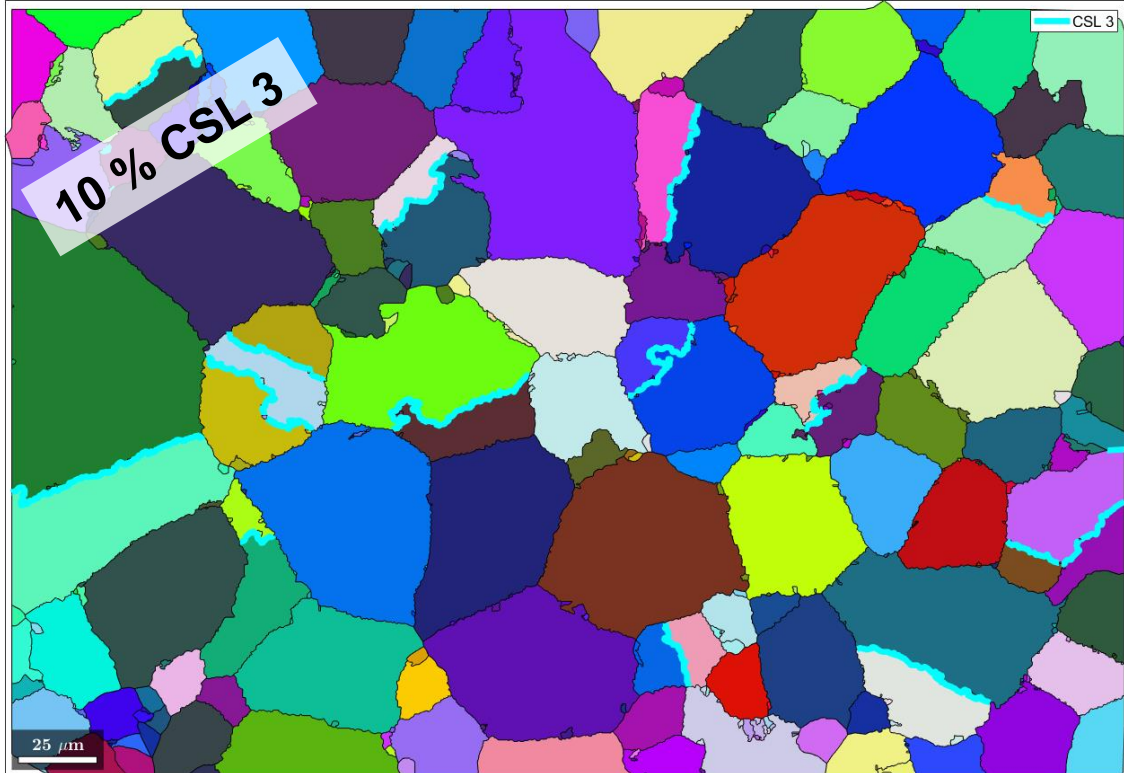
% grain reconstruction
[grains,ebstd.grainId] =
calcGrains(ebstd('indexed'),'angle',3*degree);
ebstd(grains(grains.grainSize < 3)).rotations =
quaternion.nan;
ebstd(grains(grains.grainSize < 3)).phase = 0;
ebstd(grains(grains.grainSize < 3)).grainId = 0;
[grains,ebstd.grainId] =
calcGrains(ebstd('indexed'),'angle',3*degree);
%%
job = parentGrainReconstructor(ebstd,grains) % define job
job.calcParent2Child; % get representative OR

%%
[fit,c2cPairs] = job.calcGBFit;
[gB,pairId] =
job.grains.boundary.selectByGrainId(c2cPairs);
plot(gB, fit(pairId));
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%%
job.calcGraph
job.clusterGraph
job.calcParentFromGraph
job.revert(job.grains.fit/degree > 5)
job.revert(job.grains.clusterSize < 15)

for k = 1:3 % do this three times
job.calcGBVotes('noC2C');
job.calcParentFromVote('minFit',7.5*degree)
end

```



# Kernel + growth approach (Ranger et al.)

```

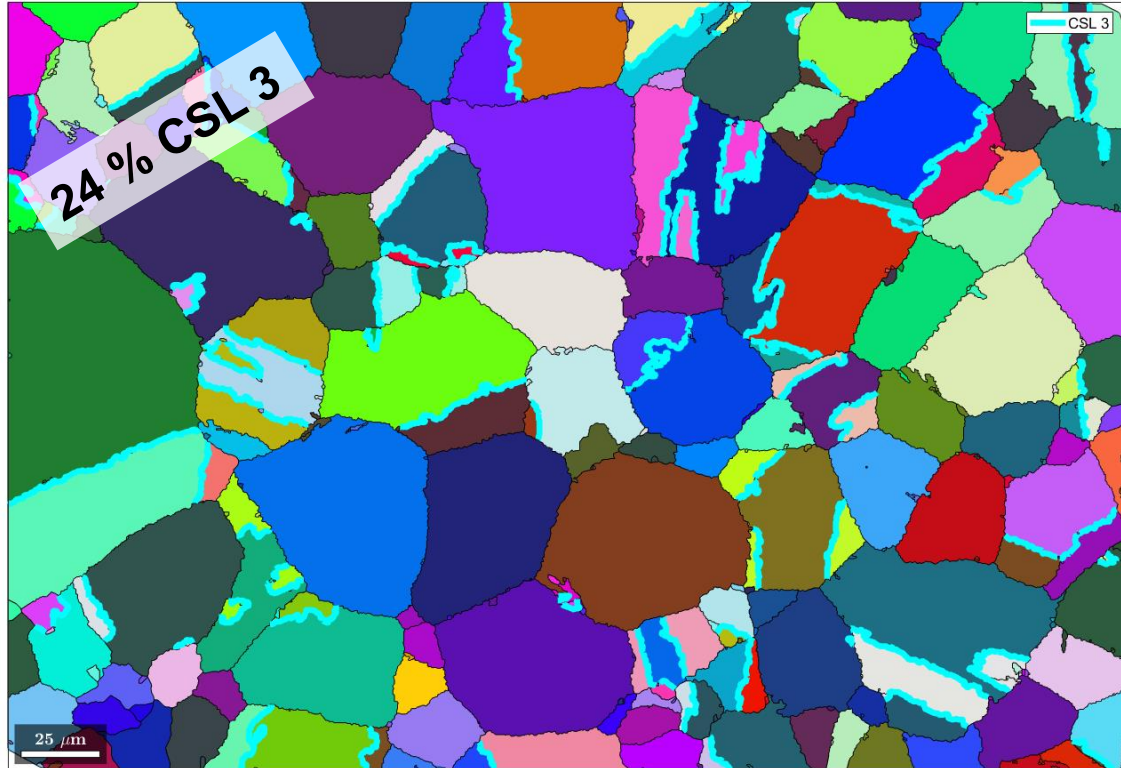
% load the data
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% grain reconstruction
[grains,ebstd.grainId] =
calcGrains(ebsd('indexed'),'angle',3*degree);
ebstd(grains(grains.grainSize < 3)).rotations =
quaternion.nan;
ebstd(grains(grains.grainSize < 3)).phase = 0;
ebstd(grains(grains.grainSize < 3)).grainId = 0;
[grains,ebstd.grainId] =
calcGrains(ebsd('indexed'),'angle',3*degree);
%%
job = parentGrainReconstructor(ebsd,grains) %
define job
job.calcParent2Child; % get representative OR

%%
job.calcGBVotes % compute votes from child grain
boundaries
job.calcParentFromVote('minVotes',4)

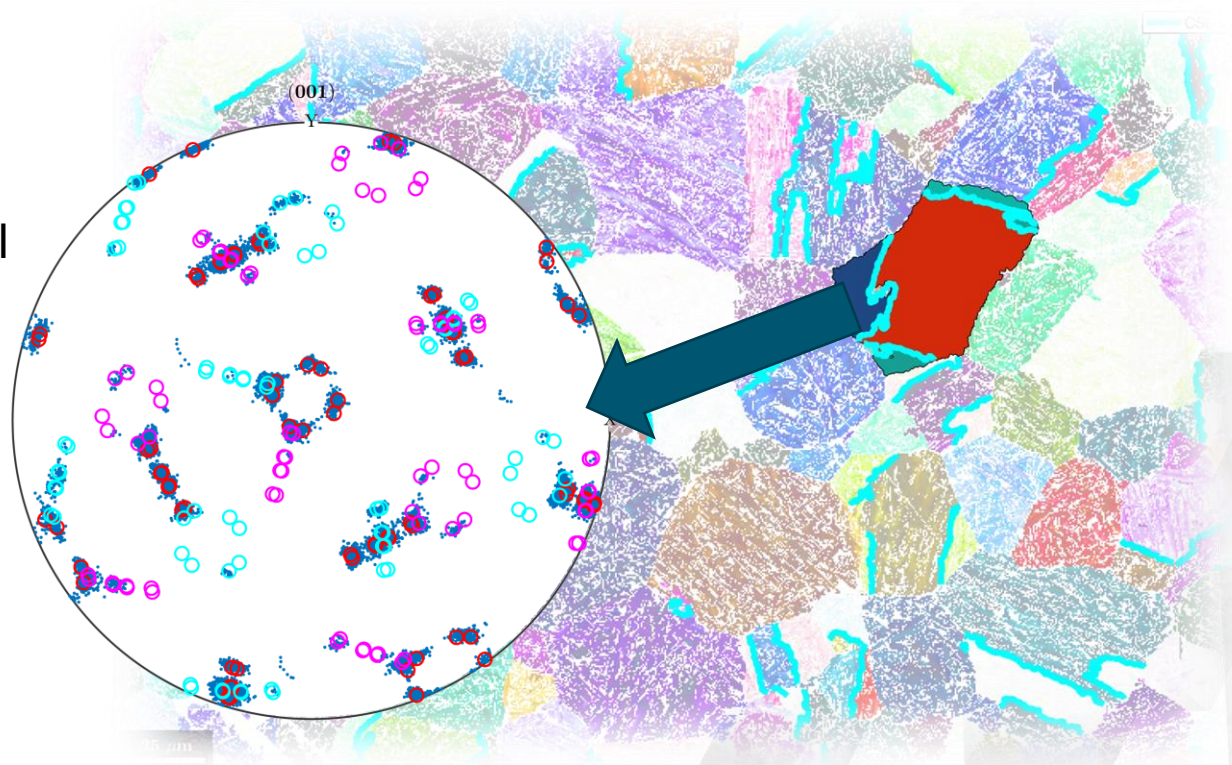
%%
for l = 1:2
job.calcGBVotes('noC2C') % compute votes ONLY from
parent-child boundaries;
job.calcParentFromVote('minFit',5*degree) %compute
parent orientations from votes
end

```

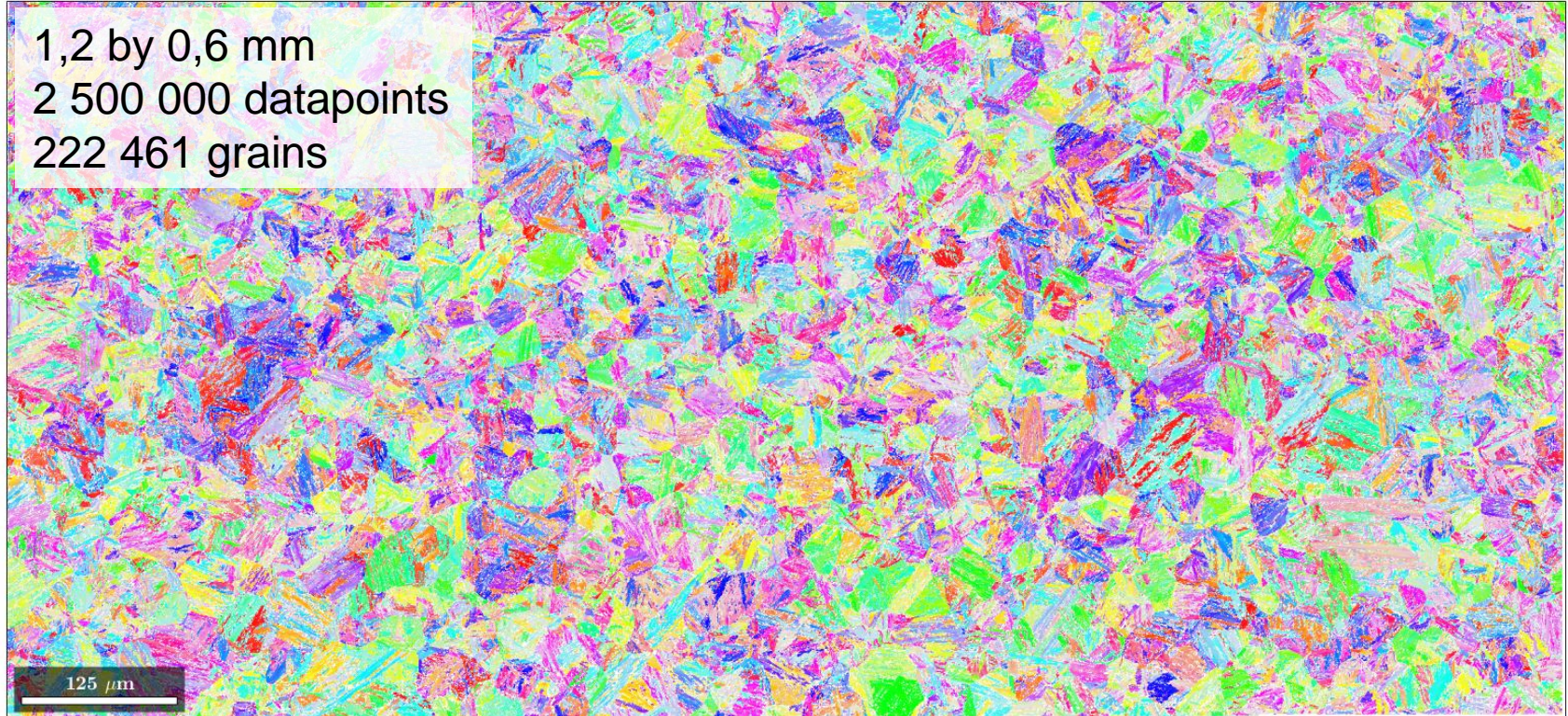


# Were all twins really indexed?

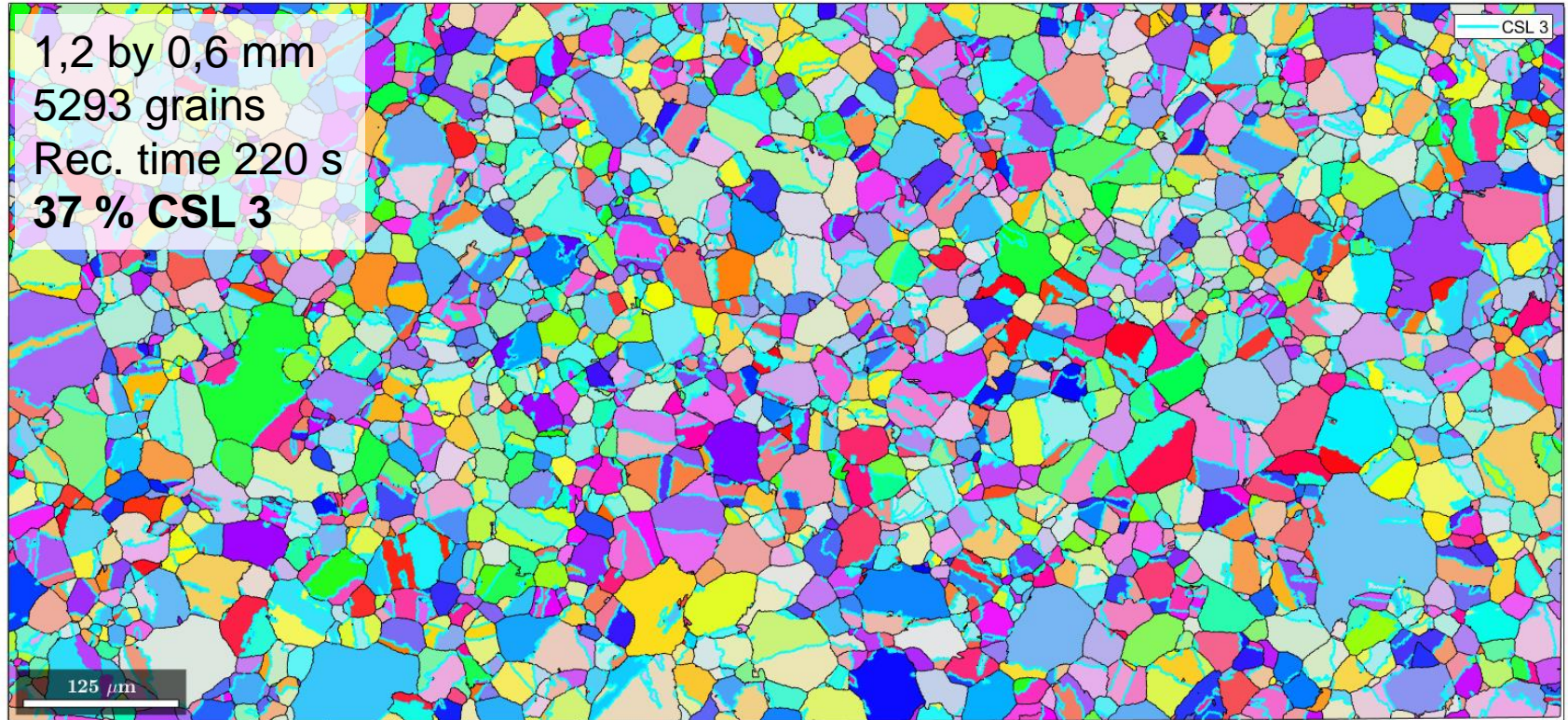
- Three twins present and identified!
- Some orientations still unaccounted for...



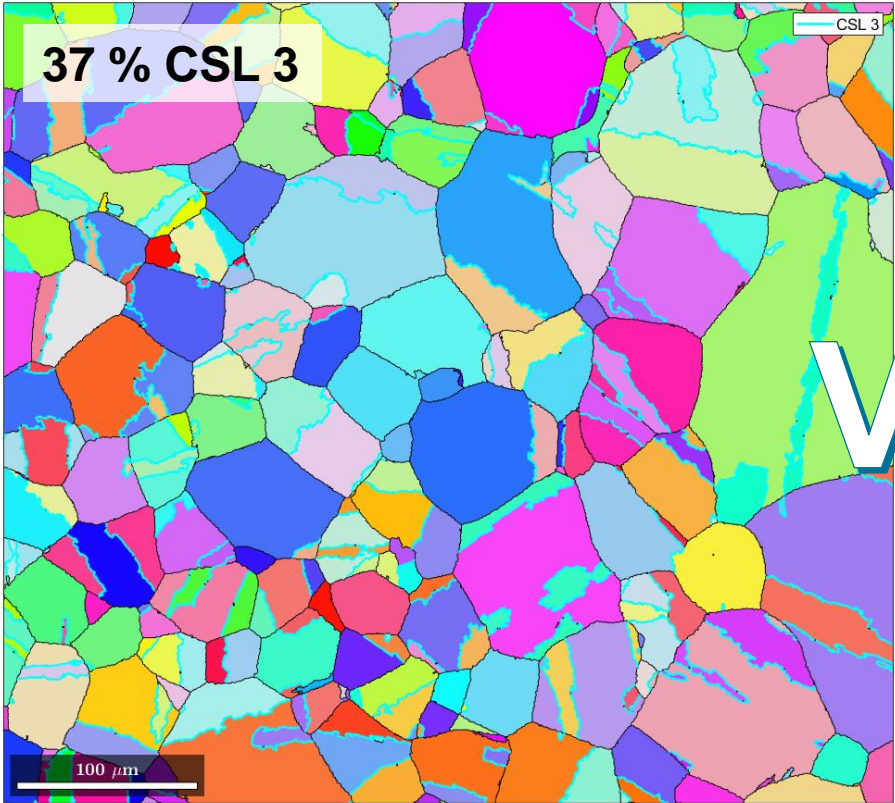
# Kernel + growth method used to reconstruct a large area map



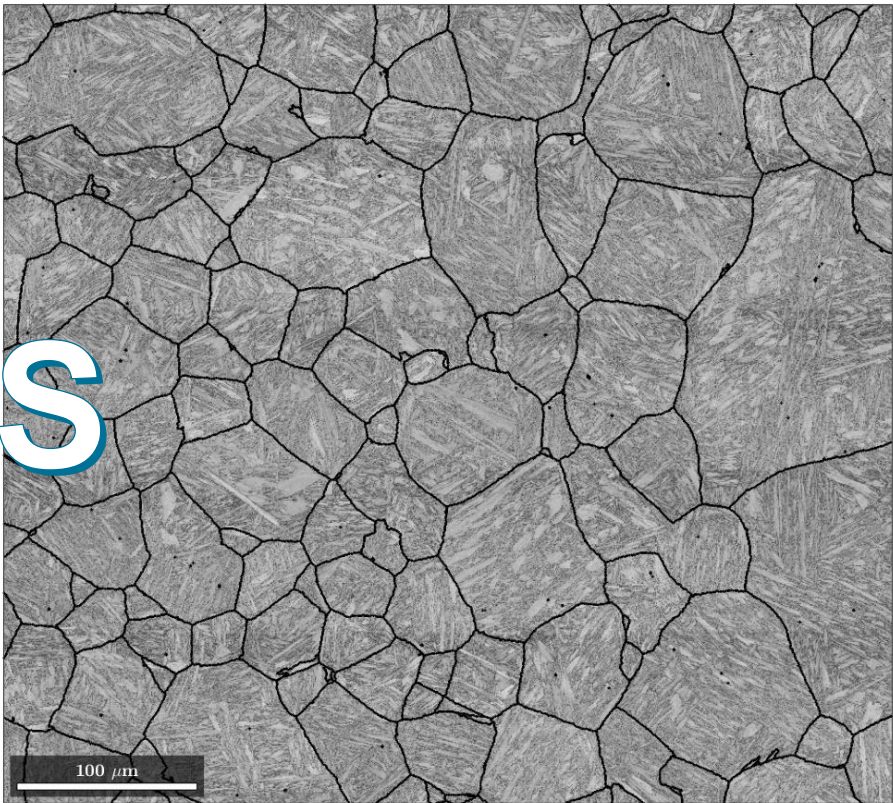
# Kernel + growth method used to reconstruct a large area map



# Twin boundaries are still challenging

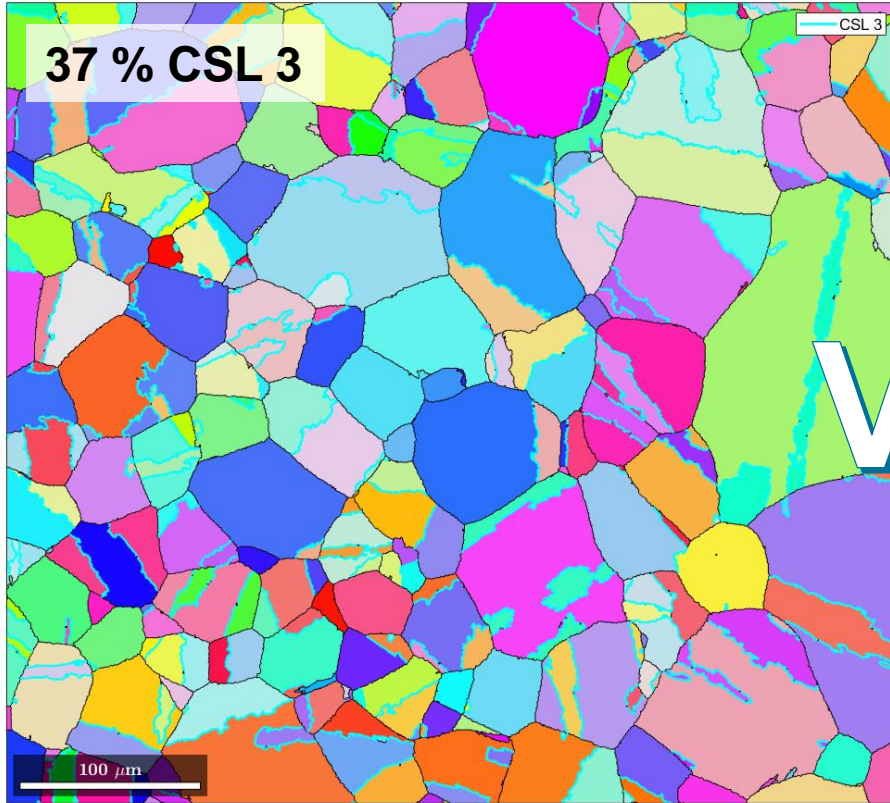


VS

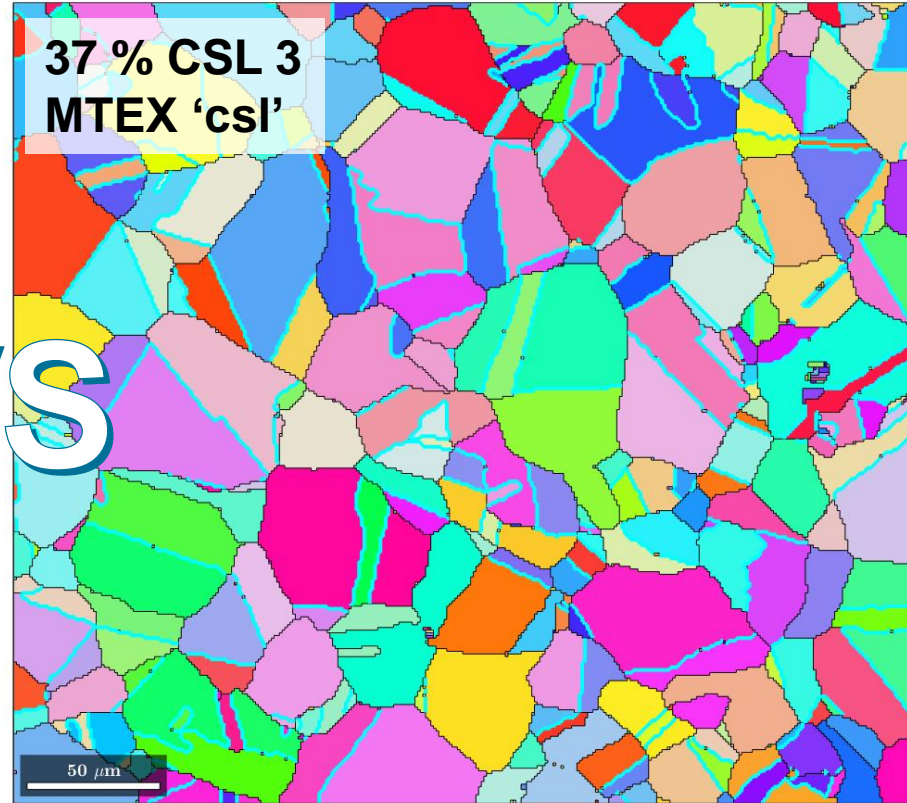




# Twin boundaries are still challenging



VS



# Conclusions

- Twin boundaries still difficult to index
- Kernel + growth approach helps a bit
- How to approach problem
  - Twin boundary does not cross packet boundary
    - Find problematic packets, test for best fit?





Tuomo Nyysönen  
[tuomo.nyysonen@swerim.se](mailto:tuomo.nyysonen@swerim.se)